

1 **Effects of halloysite content on the thermo-mechanical performances of composite bioplastics**  
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**Abstract**

The aim of this study is the design and preparation of Mater-Bi/halloysite nanocomposite materials that could be employed as bioplastics alternative to the petroleum derived products. The biocomposite materials at variable halloysite content (from 0 to 30 wt%) were prepared by using the solvent casting method. We investigated the mechanical behaviour and the thermal properties of the prepared nanocomposites in order to estimate their suitability as biocompatible packaging materials. The thermo-mechanical characteristics were correlated to the nanocomposites' morphologies, which were studied by Scanning Electron Microscopy (SEM). As a general result, the physico-chemical performances of Mater-Bi were improved by the presence of small amounts of nanotubes, which evidenced a homogenous distribution in the polymer matrix. The strongest enhancements of the thermal stability and tensile properties were achieved for Mater-Bi/halloysite 10 wt%. A further addition of nanotubes determined the worsening of both thermal stability and mechanical behaviour.

The attained knowledge represents the starting step for the development of packaging films composed by Mater-Bi and halloysite nanotubes.

## 54 **1. Introduction**

55 Environmental issues, e.g. pollution and climate change, are the most urgent challenges that science  
56 must tackle in the medium period. The need to exploit and strengthen all the possibilities that green  
57 chemistry offers is compelling. Therefore, scientists and researchers are making great efforts with  
58 the aim to provide new green tools to the society, thus ensuring a more respectful development  
59 model. Herein, the use of petroleum derived materials must be restricted due to their harmful  
60 impact on the ecosystem and their non-biocompatibility. In the most recent years, new bio-derived  
61 materials have been designed and studied with the purpose to replace traditional plastics. Hence, the  
62 term “bioplastics” derived from the need to develop novel eco-sustainable systems without waiving  
63 some important physico-chemical properties.

64 Biopolymers represent a profitable alternative for the design of new eco-friendly functional  
65 materials being completely green, eco-sustainable and non-toxic (Liu et al., 2012; Gorrasi, 2015;  
66 Biddeci et al., 2016; Rebitski et al., 2018). They can display some different features in relation with  
67 the natural source where the raw matter is extracted and used during the preparation procedure  
68 (Tharanathan, 2003; Mensitieri et al., 2011). The biopolymers charge is one of the most important  
69 feature that affect their suitability in numerous applications (Bertolino et al., 2016). Among the  
70 sustainable polymers, chitosan is positively charged, alginate and pectin are negatively charged,  
71 starch and cellulose are neutral. It is also important to consider the different  
72 hydrophilic/hydrophobic behaviours of such species that, in light of these reasons, offer a wide  
73 range of choice to material scientists and engineers.

74 Nevertheless, some limitations to the use of pure biopolymers still exist and they are related to the  
75 mechanical, thermal or gas barrier properties (Gorrasi et al., 2014; Lvov et al., 2016; Sharma et al.,  
76 2018). In order to overcome these constraints, the most promising bioplastics are prepared through  
77 the combination of both organic moieties and some inorganic fillers, such as clay nanoparticles  
78 (Dziadkowiec et al., 2017; Almeida et al., 2019; Rebitski et al., 2019). Nanoclays are attracting the

79 attention of the scientific community due to their peculiar features in terms of chemistry,  
80 morphology, aspect ratio and charge (Peyne et al., 2017; Djellali et al., 2019; Lisuzzo et al., 2019b).  
81 Among the clay nanoparticles, halloysite (Hal) is a 1:1 aluminosilicate that can be found in nature,  
82 whose main property is its distinctive hollow tubular shape combined with its eco-compatibility,  
83 no-toxicity and low cost (Joo et al., 2013; Zhang, 2017). Each halloysite nanotube is composed by a  
84 layer of tetrahedral siloxanes and a layer of octahedral aluminols forming a sheet that rolls up, thus  
85 creating two chemically different surfaces (Joussein et al., 2005). The outer surface (composed by  
86 Si-O-Si groups) is negatively charged and the inner one (composed by Al-OH groups) is positively  
87 charged in the 2-8 pH range (Lazzara et al., 2018). Moreover, halloysite nanotubes show a high  
88 aspect ratio, since the inner diameter is about 10-20 nm, the outer diameter is 50-70 nm and the  
89 length is 1-2  $\mu\text{m}$ . Within this, it should be noted that the polydispersion degree in sizes is strongly  
90 affected by the specific geological source of halloysite (Cavallaro et al., 2018a). Recently,  
91 halloysite nanotube-based functional materials have received increasing attention as evidenced by  
92 research articles (Tan et al., 2014; Zeng et al., 2019) and reviews (Yuan et al., 2015; Papoulis,  
93 2019). Halloysite is a promising nanoclay in numerous applications, including drug delivery  
94 (Viseras et al., 2008; Aguzzi et al., 2013; Dзамukova et al., 2015), catalysis (Liu et al., 2018;  
95 Sadjadi et al., 2018) and remediation (Berthonneau et al., 2015; Nyankson et al., 2015; Panchal et  
96 al., 2018; Deng et al., 2019; Wei et al., 2019) . There are many examples proving the importance of  
97 halloysite in addition to eco-compatible polymers as starting building blocks for the preparation of  
98 novel green materials with specific biomedical and technological functionalities (Silva et al., 2014;  
99 Naumenko et al., 2016; Qin et al., 2016; Liu et al., 2017; Zhou et al., 2017; Ali and Ahmed, 2018;  
100 Zhao et al., 2018; Suner et al., 2019). For instance, biocomposites with a multilayered morphology  
101 based on a layer of halloysite sandwiched between two layers of chitosan were designed for medical  
102 or fire retardancy applications (Bertolino et al., 2018). Literature reports the formation of a chitosan  
103 film with embedded clay nanotubes used for the preparation of alginate covered tablets, which  
104 exhibited pH controlled drug delivery capacity (Lisuzzo et al., 2019a). Bioplastics were prepared

105 by filling halloysite into the pectin matrix for food packaging applications (Makaremi et al., 2017)  
106 and by combining lysozyme loaded nanotubes and poly ( $\epsilon$ -caprolactone) (PCL) for the preparation  
107 of antimicrobial packaging membrane (Bugatti et al., 2017). Moreover, bionanocomposites based  
108 were obtained by the halloysite addition into cellulose and chitosan for the delivery of curcumin or  
109 tissue engineering applications, respectively (Liu et al., 2013; Huang et al., 2017).

110 In our previous work (Cavallaro et al., 2018), we investigated the effect of different nanoclays  
111 (sepiolite, halloysite, laponite and kaolinite) in the mechanical properties of Mater-Bi polymer  
112 based nanocomposites with a fixed polymer/nanofiller ratio. The results showed that halloysite is  
113 the most promising nanoclay for the preparation of bioplastics that present Mater-Bi as polymeric  
114 matrix. Here, we studied the influence of the halloysite content on both the mechanical and thermal  
115 features of the bionanocomposites in order to determine an effective protocol for the development  
116 of Mater-Bi/halloysite hybrid films with proper characteristics for packaging purposes.

117

## 118 **2. Materials and methods**

### 119 *2.1 Materials*

120 Halloysite (Hal) was supplied as a “processed product” by Imerys from their Matauri Bay  
121 operation. Mater-Bi is a Novamont product (Novara, Italy) and 1,2-Dichloroethane was purchased  
122 by Sigma-Aldrich.

### 123 *2.2 Preparation of the nanocomposites*

124 The nanocomposites based on halloysite nanotubes and Mater-Bi were prepared by solvent casting  
125 method. Firstly, the biopolymer (2 wt %) was solubilized in 1,2-Dichloroethane and Hal was added  
126 as dry powder to the solution at different concentrations, from 0 to 30% w/w. Thereafter, each  
127 mixture was magnetically stirred overnight at 25 °C and poured down into glass Petri dishes in  
128 order to evaporate the solvent, thus obtaining different nanocomposite films, which were stored in a  
129 desiccator at 25 °C.

## 130 2.3 Methods

### 131 2.3.1. Dynamic Mechanical Analysis

132 A DMA Q800 instrument (TA Instruments) was used in order to study the tensile properties of the  
133 Mater-Bi/Hal nanocomposites. In particular, the films were cut into rectangular shaped portions  
134 ( $10.00 \times 5.00 \times 0.20 \text{ mm}^3$ ) and the measurements were performed with a stress ramp of  $1 \text{ MPa min}^{-1}$   
135 at  $25.0 \pm 0.5 \text{ }^\circ\text{C}$ . The analysis of the stress vs strain curves allowed us to determine the Young  
136 modulus ( $Y_m$ ), the tensile strength ( $\sigma_r$ ) and the percentage of elongation at break ( $\varepsilon\%$ ). The  
137 integration of each curve provided the stored energy ( $E$ ) for each film due to the mechanical stress  
138 until the breaking point.

139

### 140 2.3.2. Thermogravimetry

141 Thermogravimetric analysis (TGA) was performed using a Q5000 IR apparatus (TA Instruments)  
142 under the nitrogen flow of  $25 \text{ cm}^3 \text{ min}^{-1}$  for the sample and  $10 \text{ cm}^3 \text{ min}^{-1}$  for the balance. The  
143 calibration was carried out by means of Curie temperature of standards (nickel, cobalt and their  
144 alloys) as reported in literature (Blanco et al., 2014, 2017). Each sample, whose mass was ca. 5 mg,  
145 was heated from room temperature to  $750 \text{ }^\circ\text{C}$  with a rate of  $20 \text{ }^\circ\text{C min}^{-1}$ . TGA allowed to study the  
146 thermal degradation of Mater-Bi/Hal nanocomposites and to investigate any effects on the thermal  
147 stability of these materials.

148

### 149 2.3.3. Scanning Electron Microscopy

150 The morphological features of the nanocomposites were studied by scanning electron microscopy,  
151 which was conducted using a ESEM FEI QUANTA 200F microscope. In order to avoid any  
152 charging under electron beam, each sample was coated with gold in argon by means of an Edwards  
153 Sputter Coater S150A before the analysis. The measurements were carried out in high-vacuum

154 mode ( $< 6 \times 10^{-4}$  Pa) for simultaneous secondary electrons. The energy of the beam was 20 kV,  
155 while the working distance was set at 10 mm.

156

#### 157 2.3.4. Fourier transform infrared (FTIR) spectroscopy

158 Fourier transform infrared (FTIR) measurements were performed at room temperature through a  
159 Frontier FTIR spectrometer (PerkinElmer). The spectra were recorded using 64 scans in the range  
160 between 4000 and 450  $\text{cm}^{-1}$ , while the spectral resolution was set at 2  $\text{cm}^{-1}$ . The experiments were  
161 conducted on KBr pellets with a low content ( $< 2$  wt%) of milled sample.

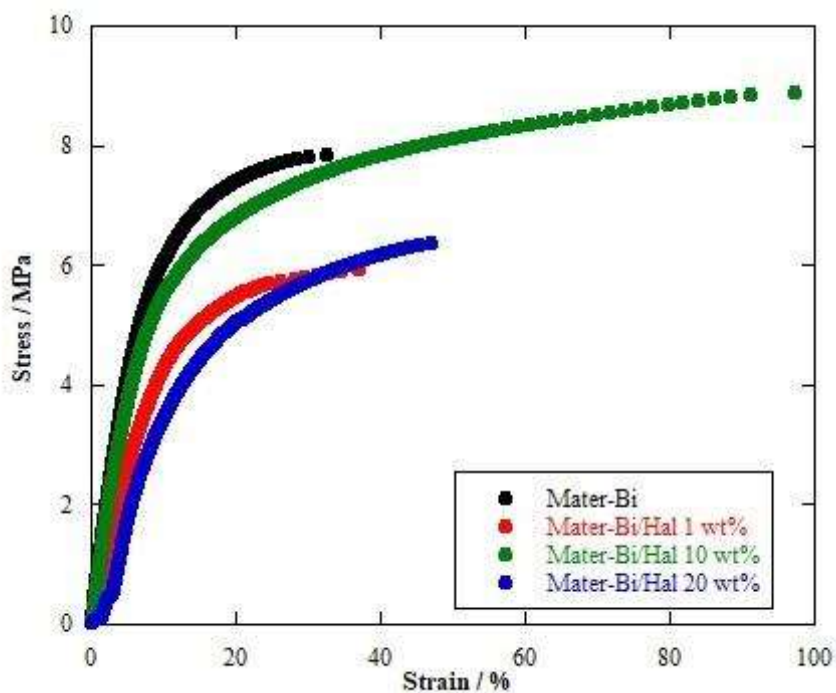
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### 163 3. Results and discussion

#### 164 3.1. Mechanical properties of Mater-Bi/Halloysite composite films

165 Some examples of stress vs. strain curves for Mater-Bi and Mater-Bi/halloysite films are presented  
166 in Figure 1.

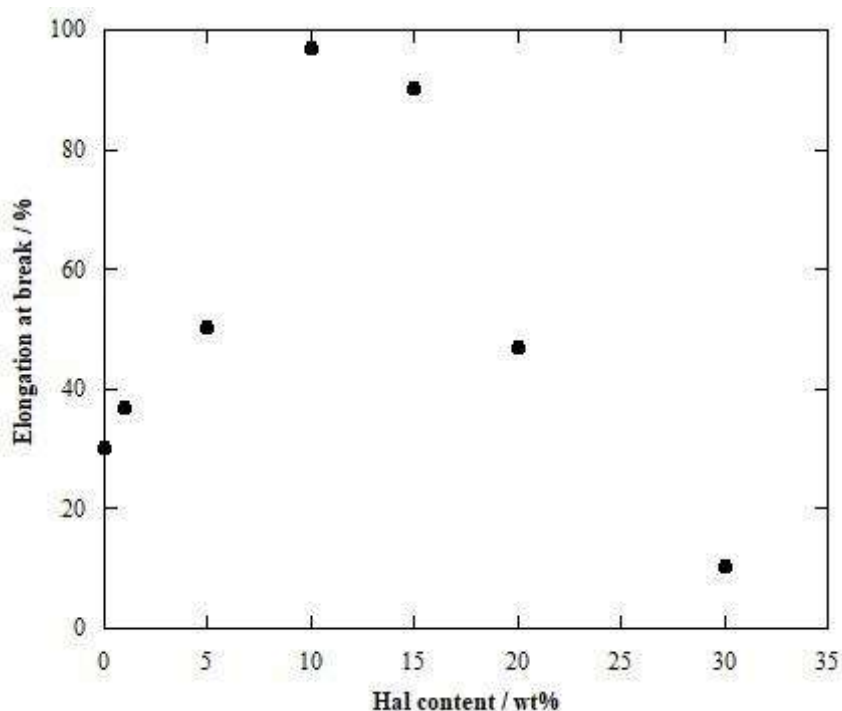
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169 **Figure 1.** Stress vs. strain curves for pure Mater-Bi and Mater-Bi/Halloysite films as a function of  
170 the nanoclay content.

171 The stress vs. strain curves (Figure 1) showed that all the prepared samples are elastic at first, since  
172 the strain increases linearly with the stress, and they convert into plastic materials when the yielding  
173 point is reached, thus the deformation becomes irreversible. Moreover, it was observed that the  
174 strain at break is much higher for the Mater-Bi/Hal 10 wt% nanocomposite in comparison with the  
175 other samples. The dependence of the elongation at break on the nanofiller content is shown in  
176 Figure 2.



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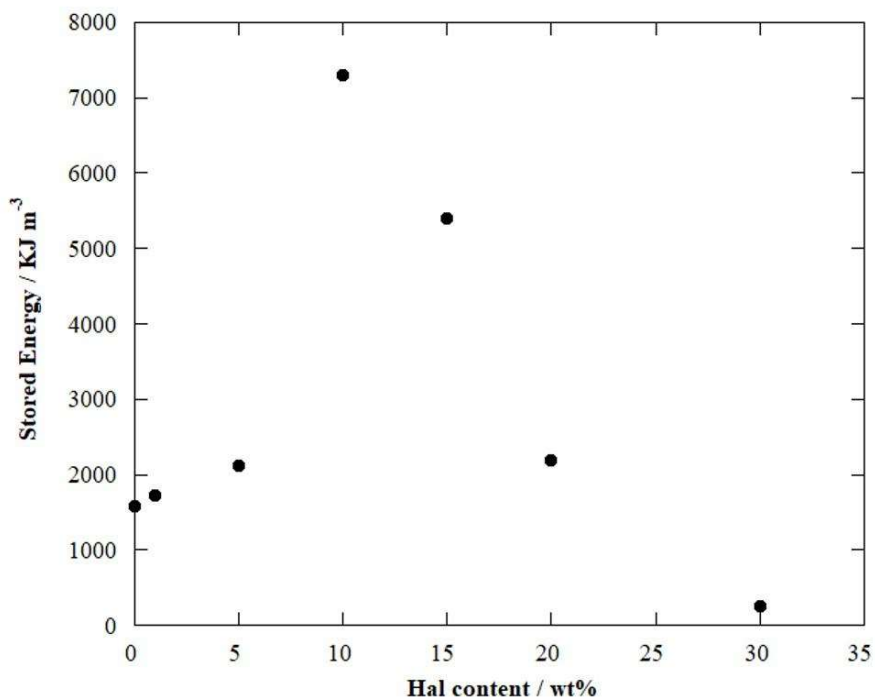
178 **Figure 2.** Elongation at break for Mater-Bi/Halloysite composite films as a function of the nanoclay  
179 content. The relative error is 2%.

180

181 It is clearly observed that the concentration of inorganic nanotubes deeply influences the ultimate  
182 elongation of Mater-Bi based films (Figure 2). In particular, we detected that the percentage of  
183 ultimate elongation is 30 and 100% for the pure Mater-Bi film and Mater-Bi/Hal 10 wt%,  
184 respectively. The further addition of nanotubes decreased the elongation at break up to ca. 10% for  
185 the composite material with the largest Hal content (30 wt%). According to the literature (Tang and  
186 Alavi, 2012; Cavallaro et al., 2013), the reduction of the maximum elongation can be explained by



187 considering the nanotubes/Mater-Bi interactions, which avoid the sliding of the polymer chains.  
188 Similar effects were observed for the stored energy of the films during the tensile experiments  
189 (Figure 3).



190

191 **Fig. 3.** Stored energy up to breaking for Mater-Bi/Halloysite composite films as a function of the  
192 nanoclays content. The relative error is 2%.

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194 Specifically, we determined that the film based on pristine Mater-Bi possess a stored energy of  
195  $1718 \text{ kJ m}^{-3}$ . This value was significantly enhanced in the composite with Hal concentration of 10  
196 wt% (ca.  $7300 \text{ kJ m}^{-3}$ ), while the Mater-Bi/Hal 30 wt% exhibited a much lower stored energy ( $254$   
197  $\text{kJ m}^{-3}$ ). These observations are in good agreement with the data of both stress at break and Young  
198 modulus (see Supporting Information).

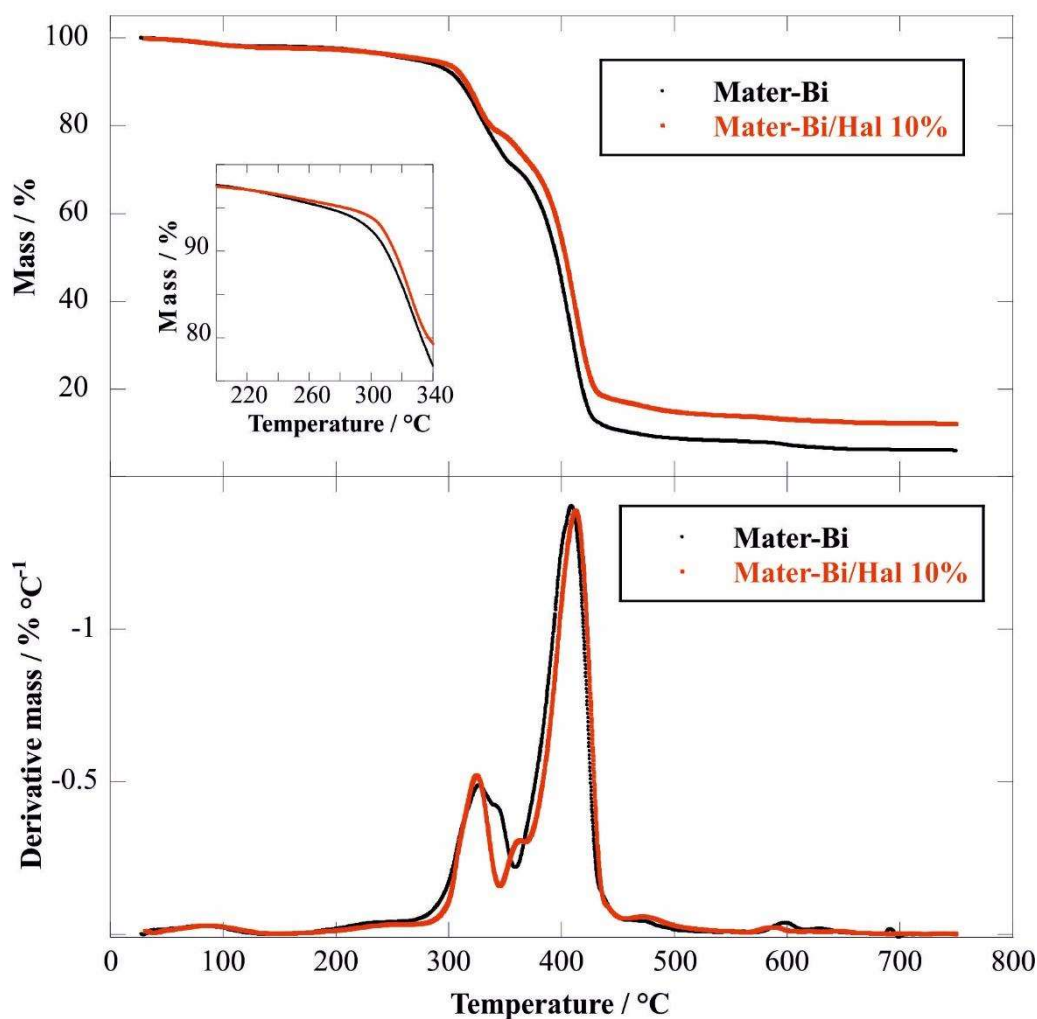
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202 3.2. Thermal properties and structure

203 Thermogravimetric analysis was aimed to study the thermal stability and the degradation properties  
204 of the Mater-Bi/Hal nanocomposite film. As examples, Figure 4 reports the thermogravimetric  
205 curves for both pure Mater-Bi and Mater-Bi/Hal\_10 wt% materials.



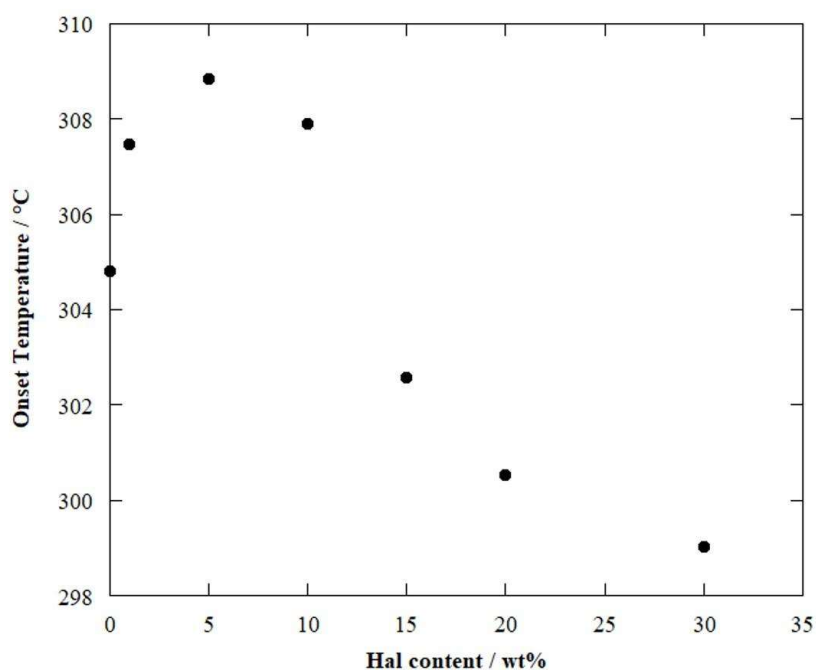
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207 **Figure 4.** Thermogravimetric (up) and differential thermogravimetric (bottom) curves for Mater-Bi  
208 and Mater-Bi/Hal 10 wt% materials. The inset shows the thermogravimetric curves for Mater-Bi  
209 and Mater-Bi/Hal 10 wt% within the temperature range between 200 and 340 °C.

210

211 Compared to pristine Mater-Bi, the nanocomposite presents a larger residual mass at high  
212 temperature as a consequence of the presence of the inorganic nanotubes (Figure 4). In this regards,  
213 the residual masses at 700 °C for all the investigated materials are presented in Supporting  
214 Information. These results evidenced that the nanocomposites with higher Hal contents possess  
10

215 larger residual masses at 700 °C. All the investigated materials showed a mass loss in the  
216 temperature range between 25 and 150 °C that can be attributed to the moisture content of the  
217 bioplastics. Both Mater-Bi and Mater-Bi/Hal 10 wt% nanocomposite exhibited two different  
218 degradation steps in the 300-400°C range as highlighted by the corresponding differential  
219 thermogravimetric curves (Figure 4). In order to explore the effect of halloysite on the thermal  
220 stability of Mater-Bi, we determined the onset temperatures from the analysis of the  
221 thermogravimetric curves. The onset temperature refers to the initial decomposition of Mater-Bi.  
222 Namely, it represents the temperature where the polymer starts to decompose. The onset  
223 temperature is obtained by the intersection of a line tangent to the baseline the with a line tangent to  
224 the inflection point of the thermogravimetric curve.



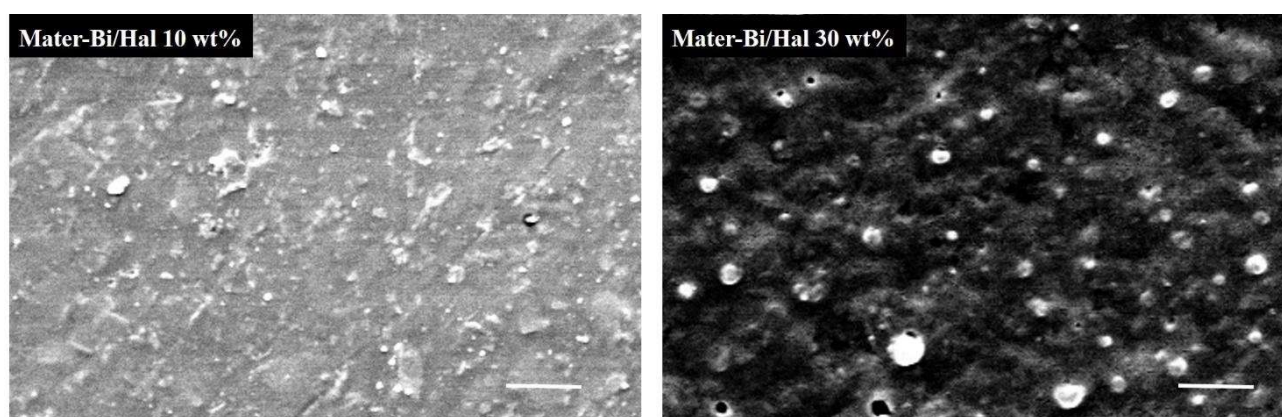
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226 **Figure 5.** Onset temperature values for pure Mater-Bi and Mater-Bi/Halloysite composite films as a  
227 function of the nanoclay content. The relative error is 2%.

228

229 Similarly to the tensile properties, the presence of small amounts of halloysite generated a slight  
230 improvement of the Mater-Bi thermal stability (Figure 5). In particular, the nanocomposite with Hal  
231 content of 10 wt% showed an enhancement of ca. 4 °C for the onset temperature with respect to that

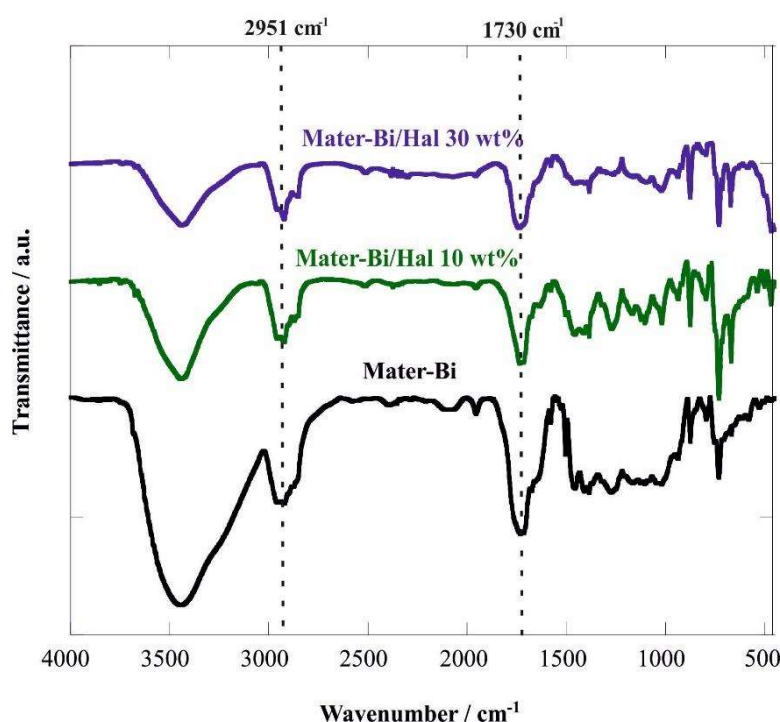
232 of pristine Mater-Bi. The slight thermal stabilization effect induced by the presence of small  
233 amounts of halloysite can be observed by the comparison of the thermogravimetric curves for  
234 Mater-Bi and Mater-Bi/Hal 10 wt% within the temperature range between 200 and 340 °C (see  
235 inset in Figure 4). Larger filler contents (> 10 wt%) induced a worsening of the polymer thermal  
236 stability as evidenced by the decrease of the onset temperature (Figure 5). Similar trends are  
237 reported for the degradation temperatures estimated by the maxima of the DTG peaks (see  
238 Supporting Information). These effects might be related to the peculiar morphological characteristics  
239 of the nanocomposites that can be influenced by their specific composition. Based on  
240 thermogravimetric results, the nanotubes should be uniformly dispersed in the polymeric matrix for  
241 the composite materials with Hal  $\leq$  10 wt%. This hypothesis was supported by SEM images (Figure  
242 6), which showed that Mater-Bi/Hal 10 wt% presents the nanotubes randomly dispersed in its  
243 surface.



244  
245 **Figure 6.** Scanning Electron Microscopy image of the Mater-Bi/Hal nanocomposites at variable  
246 composition. The scale bar is 10  $\mu$ m.

247  
248 As reported in literature (Du et al., 2006, 2010; Makaremi et al., 2017), the homogenous  
249 distribution of inorganic fillers within polymer can cause an enhancement of the thermal stability as  
250 a consequence of the barrier effect towards the volatile products of the polymer degradation.  
251 Furthermore, the encapsulation process within the Hal cavity can contribute to the improvement of  
252 the Mater-Bi resistance to the thermal degradation. On the other hand, the decrease of the onset

253 temperature at larger Hal contents could be an indication of the phase separation between polymer  
254 and nanotubes, which are not more able to create a barrier towards the volatile products of the  
255 Mater-Bi degradation. In this regards, it is reported that nanocomposites with a large Hal content  
256 exhibit a lower thermal stability compared to the corresponding pure polymers because of their  
257 layered structure (Cavallaro et al., 2011) and/or formation of nanotubes aggregates within the  
258 matrix (Cavallaro et al., 2013). As evidenced in Figure 6, Mater-Bi/Hal 30 wt% presents a rough  
259 surface with several clusters and holes indicating that the nanotubes are not homogeneously  
260 dispersed in the polymeric matrix. Accordingly, both the thermal and mechanical properties of  
261 Mater-Bi were worsened by the addition of a large amounts of halloysite nanotubes. The nature of  
262 interactions between Mater-Bi and halloysite nanotubes in the nanocomposites was investigated by  
263 FTIR spectroscopy. Figure 7 shows the FTIR spectra for Mater-Bi and Mater-Bi with variable Hal  
264 content (10 and 30 wt%).



265

266 **Figure 7.** Fourier Transform Infrared spectra for Mater-Bi, Mater-Bi/Hal 10 wt% and Mater-Bi/Hal  
267 30 wt% materials.

268 We observed that the characteristic FTIR signals of Mater-Bi are not influenced by the presence of  
269 halloysite. In particular, we focused on the peaks centered at 2951 and 1730  $\text{cm}^{-1}$ , which are related  
270 to C-H aliphatic stretching and C=O stretching vibration of polyester, respectively (Cataldi et al.,  
271 2015). Both signals did not show neither shifting nor splitting phenomena in the Mater-Bi/Hal  
272 nanocomposites highlighting that the polymer structure was not altered by the addition of the  
273 nanotubes. Based on FTIR spectra, we can state that Mater-Bi is physically adsorbed onto halloysite  
274 surfaces ruling out the presence of covalent bonds between matrix and filler.

275

#### 276 **4. Conclusions**

277 We successfully prepared a novel biohybrid material based on Mater-Bi and halloysite nanotubes  
278 by using the casting method from 1,2-Dichloroethane. The effect of the halloysite content on the  
279 tensile and thermal properties of the nanocomposites was extensively investigated by Dynamic  
280 Mechanical Analysis (DMA) and Thermogravimetry (TGA), respectively.

281 As a general result, we detected that the presence of small amounts ( $\leq 10$  wt%) of nanotubes  
282 confers improved thermo-mechanical performances with respect to those of Mater-Bi. Compared to  
283 the pure biopolymer, Mater-Bi/Hal 10 wt% composite exhibited improvements of ca. 320 and 230%  
284 for the ultimate elongation and the stored energy at the breaking point, respectively. Opposite  
285 results were detected for nanocomposites with Hal concentrations larger than 10 wt%. As example,  
286 the stored energy at break evidenced a decrease of ca. 85% in the nanocomposite with halloysite  
287 content of 30 wt%. Similar effects were observed for the elastic modulus and the stress at breaking  
288 point. As concerns the thermal behavior, we observed that the thermal stability of Mater-  
289 Bi/halloysite nanocomposites slightly depends on their specific filler content. The addition of  
290 nanotubes generated enhancements (up to 4 °C) of the polymer degradation temperature for  
291 halloysite contents lower than 10 wt%. Oppositely, a thermal destabilization of Mater-Bi was  
292 detected for nanocomposites with halloysite concentrations larger than 10 wt%. In particular,

293 Mater-Bi/halloysite 30 wt% evidenced a slight decrease (ca. 5 °C) of the polymer degradation  
294 temperature with respect to that of pure Mater-Bi that can be attributed to the formation of  
295 halloysite clusters within the matrix. According to both TGA and DMA data, we can conclude that  
296 the most promising performances were achieved for Mater-Bi/halloysite 10 wt% composite film,  
297 which might be considered as a suitable biomaterial for packaging applications.

298

## 299 **5. Supporting Information**

300 Young modulus as a function of halloysite content for Mater-Bi/Hal nanocomposites. Stress at  
301 breaking point as a function of halloysite content for Mater-Bi/Hal nanocomposites. Polymer  
302 degradation temperature (first DTG peak) as a function of halloysite content for Mater-Bi/Hal  
303 nanocomposites. Polymer degradation temperature (second DTG peak) as a function of halloysite  
304 content for Mater-Bi/Hal nanocomposites. Thermogravimetric curves and thermogravimetric  
305 parameters (residual mass at 700 °C and mass loss in temperature range between 25 and 150 °C) of  
306 all investigated Mater-Bi based materials.

307

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